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Opt electrical effects of Ag nanoparticles ink on Cerium Titanium ternary Films

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5 ABSTRACT

6 Light sensing forms a basis of fabricating solar cells which in turn promise to avail clean and 7 sustainable electrical suitable to make worthwhile contributions to solving global renewable-8 energy challenges. Current solar cell technology heavily relies on crystalline silicon wafers. 9 Due to its challenges, great research interest is now directed towards thin-film solar cells. An 10 attempt on organic semiconductors materials has presented much poorer charge transports and 11 shorter exciton diffusion lengths than inorganic ones do. These have greatly limit the thickness 12 of the photoactive layer in case of organic materials. Theoretically, materials with high 13 electrical conductivity such as metal nanoparticles are opaque. They also offer lower optical 14 transparency. Therefore, finding materials that are both transparent to visible light and 15 electrically conductive is a continues study so that today's popular devices such as liquid-16 crystal displays and organic light-emitting diodes in televisions, touch screens in phones or 17 tablet computers, electrophoretic displays in e-readers, or solar cells are visible. In this work, 18 silver nanoparticles were used to dope cerium titanium dioxide by using a laser source at low 19 pressure oxygen atmosphere irradiations. It was found that this procedure generated 20 Ce_xAg_{0.02x}TiO₂ composite when pure cerium oxide, silver metal and titanium dioxide 21 composites were used. The resulting thin films were found to possess higher absorption 22 coefficients in the UV-visible spectral region accompanied with a tunable optical band gap 23 varying between 3.42 eV to 3.78 eV. It was proposed to be suitable for wind band gap optical 24 applications like in lasers design and applications.

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26 **Keywords:** $CeTiO_2$, Composite, $Ce_xAg_{0.02x}TiO_2$ thin films, laser radiation

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28 **1. INTRODUCTION**

Silver nanoparticles have proved to have various effective applications in technology. Dispersed Ag NPs on TiO_2 thin films alone was found to enhance the photocurrent generation with five folds as compared to TiO_2 under UV light irradiation [2]. Silver modifies many nanocomposites under visible light irradiation for complete inhibition of microorganism activities. Cerium titanium dioxide trioxide has shown some useful functional properties promising technological applications especially in gas sensor technology and electrochromic devices [5]. 35 By including metal nanoparticles silver or gold in a gas sensor [13], it is found to modify in 36 their physical, chemical and structural properties. It has been found that gas sensor activity 37 increase three times higher when loaded Ag nanoparticles under visible-light irradiation. It has 38 also been found to effective enhance sensitivity and selectivity electrochromic devices [7]. 39 They have also been proposed as effective in reducing the response [5] and recovery times in 40 gas sensors [8, 15]. It is applications include removing biological impurities from drinking 41 underground water supplies, modifying nanostructures through different processing methods 42 like sol-gel procedures, magnetron sputtering [9], direct femtosecond laser irradiation or pulsed 43 laser deposition [10]. In this study, pulsed laser deposition technique was used and we present 44 results on the effect of silver nanoparticles CeTiO₂ thin films.

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46 **2. METHODOLOGY**

47 **2.1 Cleaning of Substrates**

48 The SiO_2 quartz substrates were cleaned with acetone in a ultrasonic bath. The target was 49 thoroughly cleaned before any preliminary trials were done and also after the trials.

50 2.2 Preparation of Vacuum Chamber

51 Prior to each irradiation the vacuum chamber was evacuated down to a residual pressure of 10^{-4}

Pa. This pressure was maintained constant during the thin films synthesis process. High purity
oxygen (99.9%) was then circulated inside the irradiation chamber through a calibrated gas
inlet.

55 2.3 Reagents and Chemicals

56 The chemicals and reagents included cerium, titanium, silver, silicon substrate, high purity 57 oxygen (99.9%) among other laboratory chemicals and apparatus.

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59 **2.4. Growth procedures**

The growth of silver doped CeTiO₂ thin films was performed inside a stainless steel reaction chamber with a pulsed frequency laser fixed at $\lambda = 255$ nm, $\tau_{WHM} \sim 8$ ns, v = 20 Hz; laser fluency fixed at 5 J/cm² at 0.1 MPa pressure. Silver nano-particle concentration was varied between 1.5 to 3 wt.% range and sintered at 1000 °C for 2 hours. The substrates were positioned at a separation distance 30 mm from the target surface and parallel to it. The substrate temperature was fixed at about 630 °C during the film growth and only 10,000 laser pulses were applied to each thin film. The films were cooled down with a ramp of about $10 \, {}^{0}\text{C}$ per minute at 0.5 MPa that used during the deposition experiments.

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69 **2.5 Thin film characterization**

The optical measurements were performed with a double beam Perkin Elmer Lambda 19 spectrophotometer in the wavelength range of 300nm to 1100 nm while compositional and crystallinity status were investigated by XRD and EXRD measurements by configuring the Xray diffraction in θ -2 θ configuration with a Philips MRD diffractometer (CuK α , λ =1.5418 Å radiation).

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76 **3.0 RESULTS**

77 **3.1 Elemental Composition of CeTiO₂**

The obtained SAED pattern resulted into inter-planar distances of 0.385 and 0.192 nm assigned to the (001) and (002) lattice plane reflections of the orthorhombic CeTiO₂ phase [4,7], with lattice parameters a = 7.318 Å, b = 7.451 Å, and c = 3.684 Å, as referred in the JC-PDS 20-1324. Hence, there was no indication of the presence of silver or silver oxide observed.

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83 **3.2 Structural Properties of CeTiO₂ Thin films**

The crystalline structure of the thin films was investigated using the XRD technology. The observed diffractogram curves composed of diffraction lines at 23.1° , 47.2° and 50.4° attributed to the (001), (002) and (112) lattice plane reflections. This implied an orthorhombic phase with a preferred orientation along the [001] crystal direction. With increase in Ag concentration, the intensity of the 23.1° line corresponding to (001) lattice plane reflection decreased gradually with full width at half maximum increasing [9].

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91 **3.2.1 Structural Properties silver doped of CeTiO₂ Thin films**

Adding silver nano-particles to the film, the XRD peaks were modified. It was attributed the line 23.6° that appeared at (020) lattice plane reflection. About 0.2° degree shift towards higher values on the line was observed that corresponded to (001) lattice plane reflection. This was also attributed to the slight change of the d-spacing lattice parallel to the surface of substrate. It was concluded that silver nanoparticles as a dopant lead to an increase in the average dimensions crystals. The average size of nanocrystallites in the films was determined by the Scherrer equation [10]:

99
$$D_{hkl} = \frac{0.9\lambda}{\beta_{hkl}\cos\theta_{hkl}}$$
(1.1)

100 where λ is the X-ray wavelength, θ_{hkl} is the Bragg diffraction angle and β_{hkl} is the full width at 101 half-maximum.

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103 **3.3. Optical properties**

104 **3.3.1 Chromaticity coordinates and tristimulus**

It is known that chromaticity coordinates are used to specify the colour observed according to the CIE chromaticity diagram. Equally, standard tristimulus value is used as a measure of the brightness of a surface or colour. By using Pyton 2.7 program [1, 2, 9], optical transmittance spectra, chromaticity coordinates and standard tristimulus value were calculated and tabulated as shown in table I and it is graphically represented by figure 1. The dark blue colour was attributed to oxygen deficiency in the thin films. This further implied that CeTiO₂ underwent partial decomposition at the target as it was being irradiated by the subsequent laser pulses.

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Table I: Chromaticity coordinates and tristimulus of doped thin films

| Ag conc. | Chromaticity | Tristimulus | Colour | | | |
|----------|--------------------|-------------|----------------------|--|--|--|
| wt.%] | coordinates (x, y) | Y | Status | | | |
| 3.0 | (0.35, 0.34) | 817 | Dark blue | | | |
| 2.5 | (0.29, 0.32) | 321 | slightly Transparent | | | |
| 2.0 | (0.23, 0.16) | 19 | Fairly transparent | | | |
| 1.5 | (0.18, 0.16) | 9 | Transparent | | | |
| 1.0 | (0.14, 0.13) | 3 | Very transparent | | | |
| 0.5 | (0.12, 0.11) | 1 | Very transparent | | | |
| 0.0 | (0.11, 0.1) | 1 | Very transparent | | | |



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Figure 1: Tristimulus of doped thin films

116 **3.3.2 Optical Transmittance**

117 The average transmittance of the un-doped CeTiO₂ film was close to 90 % in the visible 118 spectral range while the one containing silver (Ag) had an average decrease of greater than 20 119 %. This was attributed to the formation of oxygen vacancies at low supply oxygen [3] since this low supply pressure does not stabilize stroichiometric formation of CeTiO₂ crystals. 120

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122 3.3.3 Optical Absorption

123 Optical absorption increased as Ag concentration increased. This was attributed to the surface 124 plasmon resonance absorption resulting from the embedded Ag nano-particles as the absorption 125 was observed to shift towards higher wavelengths as silver concentration increased. SPR peak 126 positions below 410 nm were observed at very low Ag nano-particle concentrations and thus 127 optical absorption coefficient, α , was calculated from [4];

128
$$\alpha = A \ln \frac{10}{d} \tag{1.2}$$

129 where *A* stands for the absorbance and *d* for the thickness of the films.

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131 3.3.4 Determination of thickness

132 Using surface profilometry data, the average thickness of the un-doped and doped thin films 133 was evaluated to be around 98 nm.

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135 3.3.5 Optical Band Gap, Eg

136 Fundamentally absorption coefficient follows the relation in Eq. 4 where m = 2 for the indirect

137 transitions and $\frac{1}{2}$ for directly allowed transitions [9];

138
$$\alpha hv \sim (hv - E)$$

$$\alpha hv \sim (hv - E_g)^m \tag{3}$$

The Scout Software was used determine the possible transitions by simulating $(\alpha h \nu)^{1/m}$ against 139

140 hv, photon energy for m = 2 and for $m = \frac{1}{2}$. Table II shows the obtained band gaps.

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| CeTiO ₂ | Ce _x Ag _{0.02x} TiO ₂ |
|--------------------|--|
| 2.66 | 3.42 |
| 2.78 | 3.55 |
| 2.87 | 3.76 |
| 3.03 | 3.79 |

Table II: Band gaps for Thin films

142

143 This was similar to determining the band gap from plots of Eq. 4 for direct band gap values can 144 be estimated [10];

145

$$(\alpha h v)^2 \sim h v - E_g \tag{4}$$

146 Band structure calculations were done on CeTiO₂ thin films (reported elsewhere- Mosiori et al., 2016) revealed oxygen deficiency. The narrow changes observed in band gaps was 147 148 attributed to charge carrier transfer from Ag to CeTiO₂ layer that might have resulted into a 149 downward shift of the conduction band and at the same time, an upward shift of the valence 150 band hence increasing the band gap, E_g.

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152 3.3.6 Urbach energy, E₀

153 As observed in Mosiori et al. 2016, at photon energy hv, below the absorption edge, it is 154 possible to evaluate the Urbach energy or the Urbach tail. This is because the structural 155 disorders observed generate band tails with localised states and therefore the Urbach energy, 156 E_0 , can easily be estimated from [5];

 $\alpha(hv) = \alpha_{c} e^{(\frac{hv}{E_{o}})}$ 157 (1.3)

This means that the dependence of Urbach energy on Ag concentration may introduce some 158 159 lattice disorder in the composite material.

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 Table III: Average size and Urbach Energy

| Ag concentration | Nano-crystallites | Urbach energy | | | |
|------------------|-------------------|---------------|--|--|--|
| [wt.%] | size [nm] | [meV] | | | |

UNDER PEER REVIEW

| 3.0 | 85 | 297 |
|-----|----|-----|
| 2.5 | 63 | 361 |
| 2.0 | 51 | 398 |
| 1.5 | 41 | 421 |
| 1.0 | 33 | 441 |
| 0.5 | 29 | 496 |
| 0.0 | 17 | 573 |

162 If this expression is used to plot a curve, then the inverse of the slope will be a linear fit to the 163 logarithmic plot of the absorption coefficient [7]. The value of Urbach energy calculated in this 164 work as shown in table III and further illustrated by figure 2 was fairly similar for undoped 165 CeTiO₂ thin films as reported in literature elsewhere by other growth methods [3, 6, 8,11].



Figure 2: Relationship between average Silver particle and Urbach energy

3.4 Electrical properties

3.4.1 Current-Voltage curve by CSAFM on CeTiO₂ thin films

183 CSAFM technique was used to determine the local conductivity of $CeTiO_2$ thin films using 184 biased Pt-Ir coated silicon tips in contact mode. All the curves were non-ohmic as depicted by

table IV. It was observed that the turn-on voltage value ranged between 0.1V to 0.3V while

186 their corresponding resistance at 0 V range from about 0.4 G Ω to 2.5 G Ω . The small element of

- conduction was attributed to the semiconducting properties [12] of CeTiO₂ thin films. 187
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189 3.4.1 Current-Voltage curve by CSAFM on Ag doped CeTiO₂ thin films

190 Almost similar I-V curves were obtained for Ag doped CeTiO₂ thin films but the curves were 191 slightly or nearly-ohmic in character. The slight deviation from the linearity was attributed to 192 the contribution of the semiconducting behaviour [11] of CeTiO₂ while the ohmic behaviour 193 was sttributed to the presence of evenly dispersion Ag nanoparticles present. From table IV and 194 as depicted in figure 2, the gradual transition behaviour of I-V curve is a property that can be 195 used in sensor applications to ensure better selectivity and sensitivity.

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Table IV- Current-Voltage curve by CSAFM technique.

| | Thin Film | CSAFM Measurements | | | | | | | | | | | |
|-----|----------------------|--------------------|-------|------------|----------|------|------|-----|---------|-----|-----|-----|-----|
| | | Voltage | -0.5 | -0.4 | -0.3 | -0.2 | -0.1 | 00 | 0.1 | 0.2 | 0.3 | 0.4 | 0.5 |
| | | (µV) | | | | | | | | | | | |
| | CeTiO ₂ | Current | -6 | -4 | -2 | -1.5 | -0.1 | 0.0 | 0.1 | 1.1 | 2.1 | 3.9 | 7.3 |
| | | (<i>n</i> A) | | | | | | | | | | | |
| | CeAgTiO ₂ | Current | - | - | -6.1 | -4 | -1 | 0.0 | 1.9 | 3.9 | 7.1 | - | - |
| | | (<i>n</i> A) | | | | | | | | | | | |
| 198 | | I | | | | | | | | I | I | | |
| 199 | | | | | | | | | - CeTiO | 2 | | | |
| 200 | | ٤ | 3-] | | | | | _ | - CeAgT | īO2 | | | |
| 201 | | e | - | | | • | 、 | | | | | | |
| 202 | | 4 | + - | | | | | | | | | | |
| 203 | | (Yu | 2 - | | | | | | | | | | |
| 204 | |) (|) - | | | • | | | | | | | |
| 205 | | Curr | 2 - | _ | - | | | | | | | | |
| 206 | | -4 | ;]/ | , - | , | | | | | | | | |
| 207 | | -8 | -0.6 | -0.4 | -0.2 | | 0.2 | 0.4 | 0.6 | | | | |

208

209 Figure 4: Comparison of I-V curve by CSAFM technique for CeTiO₂ aand CeAgTiO₂.

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Voltage (µV)

212 Conclusions

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- 213 A nano-composite films of $Ce_xAg_{0.02x}TiO_2$ were grown by pulsed laser deposition. The effect of 214 Ag concentration on crystalline status and optical properties was investigated. It was observed 215 that there was a gradual band gap increase. This was attributed to the thin film's change in chemical composition as Ag was introduced. Electrical properties portrayed promising features 216 217 for the design of new composite materials for applications electrochromic devices and gas 218 sensors. 219 220 221 References 222 1. Ritala, M., Leskelä, M., Nykänen, E., Soininen, P., & Niinistö, L. (1993). Growth of 223 titanium dioxide thin films by atomic layer epitaxy. Thin Solid Films, 225(1), 288-295. 224 225 2. Arabatzis, I. M., Stergiopoulos, T., Bernard, M. C., Labou, D., Neophytides, S. G., & Falaras, P. (2003). Silver-modified titanium dioxide thin films for efficient 226 photodegradation of methyl orange. Applied Catalysis B: Environmental, 42(2), 187-227 201. 228 229 3. Machida, M., Norimoto, K., & Kimura, T. (2005). Antibacterial activity of 230 photocatalytic titanium dioxide thin films with photodeposited silver on the surface of 231 sanitary ware. Journal of the American Ceramic society, 88(1), 95-100. 232 233 234 4. Hass, G. (1952). Preparation, properties and optical applications of thin films of 235 titanium dioxide. Vacuum, 2(4), 331-345. 236 237 5. Justicia, I., Ordejón, P., Canto, G., Mozos, J. L., Fraxedas, J., Battiston, G. A., ... & Figueras, A. (2002). Designed Self-Doped Titanium Oxide Thin Films for Efficient 238 Visible-Light Photocatalysis. Advanced Materials, 14(19), 1399-1402. 239 240 241 6. Argall, F. (1968). Switching phenomena in titanium oxide thin films. Solid-State *Electronics*, 11(5), 535-541. 242 243 244 7. Sberveglieri, G., Comini, E., Faglia, G., Atashbar, M. Z., & Wlodarski, W. (2000). Titanium dioxide thin films prepared for alcohol microsensor applications. Sensors and 245 246 Actuators B: Chemical, 66(1), 139-141. 247 248 8. Suzuki, T., Kosacki, I., Anderson, H. U., & Colomban, P. (2001). Electrical 249 conductivity and lattice defects in nanocrystalline cerium oxide thin films. Journal of 250 the American Ceramic Society, 84(9), 2007-2014.
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