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5 ABSTRACT

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We synthesized a quasi-two-dimensional Heisenberg antiferromagnet on a square lattice formed in α -RbCrF₄ by improving the pretreatment method before primary sintering. X-ray diffraction measurements revealed peak splitting, indicating the presence of a superstructure in the ab-plane due to the distorted square lattice caused by the tilting of CrF_6 octahedra; however, the structure of α -RbCrF₄ was not determined. Magnetic susceptibility measurements revealed a broad peak at $T_{max} \approx 47 \text{ K}$ and an antiferromagnetic phase transition at T_N = 29.3(2) K. Following several previous theoretical investigations, we estimated intra-layer ($J_{\rm intra}$) and inter-layer ($J_{\rm inter}$) exchange interactions to be $J_{\rm intra}$ / $k_{\rm B}$ = –6.6(1) K and $J_{\text{inter}}/J_{\text{intra}} \approx 0.05$, respectively. As a result, we found that α -RbCrF₄ is a quasi-two-dimensional Heisenberg antiferromagnet.

Magnetic properties of a quasi-two-dimensional

Heisenberg antiferromagnet α -RbCrF₄

Original Research Article

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Keywords: Quasi-two-dimensional magnet, X-ray diffraction, magnetic susceptibility, α -RbCrF₄

9 10 **1. INTRODUCTION**

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12 Recently, magnetoelectric multiferroic materials have received much attention because of the possibility 13 of either "magnetic control of ferroelectric domains" or "electric control of magnetic domains"[1,2,3]. In 14 ferromagnetics and ferroelectrics, the switching from one domain orientation to another occurs because 15 of the application of external perturbation, which changes the preferred, lowest energy orientation of the 16 order parameter from one state to another. In addition to magnetic and electric fields, mechanical stress 17 can have a switching effect in ferroelastic materials. In addition to magnetic and electric fields, 18 mechanical stress can have a switching effect in ferroelastic materials. Using point groups of prototypic 19 and ferroic phases, Aizu classified the cases where ferromagneticity, ferroelectricity, and ferroelasticity 20 coexist and completely couple with each other[4].

Many investigations have been reported for successive structural phase transitions in $A^{I}M^{III}F_{4}$ 21 compounds, where the $M^{III}F_6$ octahedra are centered in a square-based parallelepiped of A^+ cations in 22 the so-called TIAIF₄-type structure[5]. Figure 1 shows an aristotype $A^{I}M^{III}F_{4}$ structure. The corner-23 sharing $M^{III}F_6$ octahedra result in a square lattice with each layer separated by A^{I} cations, leading to a 24 good two-dimensionality in A^IM^{III}F₄ compounds. For example, in non-magnetic compounds TIAIF₄ and 25 RbAIF₄[6.7], internal strains were investigated and the switching of ferroelastic domains by uniaxial stress 26 27 was demonstrated, although a ferroelastic-ferroelectric effect can not be expected because of the non-28 polar space group of these materials. Furthermore, in magnetic compounds with S = 5/2, RbFeF₄ and CsFeF₄[8,9,10,11], an orthorhombic (mmm)-tetragonal (4/mmm) structural phase transition causes 29 30 spontaneous strain, and then, an antiferromagnetic phase transition occurs far below the structural phase transition temperature. Temperature (T) dependence of magnetic susceptibilities (χ) of RbFeF₄ and 31 CsFeF₄ shows a typical two-dimensional antiferromagnetic behavior. On the other hand, in magnetic 32 33 compound with S = 1, CsVF₄[12,13], a sharp $\chi(T)$ peak corresponding to an antiferromagnetic phase 34 transition appeared at a magnetic field (H) of 200 Oe, and the $\chi(T)$ curves for field cooling (FC) and 35 zero-field cooling (ZFC) overlapped. At $H \ge 5 \text{ kOe}$, a ferromagnetic moment was induced by the magnetic 36 field and the splitting of $\chi(T)$ curves for the FC and ZFC appeared. Therefore, we hope that mechanical

37 stress can be used to switch the magnetic and/or ferroelectric domains in ferroelastic $A^{I}M^{III}F_{4}$ 38 compounds.

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Previously, we have intensively studied a series of chromium fluorides, $A^{I}CrF_{4}$ ($A^{I} = K$ and Cs), because 40 41 of their highly frustrated magnetic structures such as triangular spin tubes[14,15]. Table 1 presents the 42 structural phase diagram of A^{I} CrF₄ ($A^{I} = K$, Rb, Cs), as previously reported by Kozak[16]. In equilateral 43 triangular spin tube $CsCrF_4$, no structural isomer exists below the melting point. However, a structural 44 isomer was observed in KCrF₄ and RbCrF₄ when the sintering temperature was varied. In KCrF₄, non-45 equilateral triangular spin tube α -KCrF₄ was crystallized below 768 °C [15], whereas β -KCrF₄ consisting of the CsCrF₄-type structure was crystallized above 768 °C. In RbCrF₄, α -RbCrF₄ consisting of a TIAIF₄-46 47 type structure was crystallized below 750 °C [17], whereas β -RbCrF₄, consisting of a CsCrF₄-type structure, is crystallized above 750 °C. The magnetic properties of RbCrF₄ were studied without any 48 49 distinction between the α - and β -phases[18,19].

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51 We have previously observed that the magnetic susceptibility of CsCrF₄ is strongly affected by a small 52 amount of paramagnetic impurities and/or imperfect crystallization[14,15]. A crystallization method that 53 enables the synthesis of high-quality α -RbCrF₄ must be developed to confirm the magnetic ground state. 54 Figure 1 shows the aristotype structure of α -RbCrF₄, which we believe a quasi-two-dimensional 55 Heisenberg antiferromagnet with S = 3/2. At room temperature, the superstructure in the *ab*-plane arising from the distorted square lattice caused by the tilting of CrF₆ octahedra has been reported[17]. In this 56 study, we obtained highly crystalline α -RbCrF₄ and performed X-ray diffraction (XRD) and magnetic 57 susceptibility measurements to confirm that high-quality samples were prepared and to determine the 58 59 heretofore unreported structural and magnetic properties of α -RbCrF₄.

Table 1: Structural phase diagram of $A^{I}CrF_{4}$ ($A^{I} = K$, Rb, and Cs) in Ref. 16.

	lpha - phase (low-T phase)	Critical Temperature (°C)	eta - phase (high- T phase)
KCrF₄	non-equilateral triangular spin tube	768	CsCrF₄-type
RbCrF₄	square lattice (TIAIF₄-type)	750	CsCrF₄-type
CsCrF₄	equilateral triangular spin tube	No structural isomer below melting point.	

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Figure 1: Schematic of aristotype structure of $A^{I}M^{III}F_{4}$ compound. Each layer is stacked without translation in the *ab*-plane. In α -RbCrF₄, A^{I} and M^{III} correspond to Rb⁺ and Cr³⁺, respectively.

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66 2. SAMPLE PREPARATION

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We prepared polycrystalline samples of α -RbCrF₄ using a method similar to that employed in the synthesis of high-quality CsCrF₄, i.e., using a conventional solid-state reaction method[15]. We mixed the RbF and CrF₃·4H₂O starting materials in accordance with the stoichiometry and then heated them at 200 °C for more than 48 h under vacuum with $P < 1 \times 10^{-3}$ Pa to dehydrate the crystals. Sintering was then performed at various temperatures below 750 °C. The final sample color was dark green. To further purify the samples, we improved the pretreatment method before primary sintering, as discussed in the next section.

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76 To examine the sample crystal structure and phase purity, we performed powder XRD measurements at 77 room temperature. The XRD data were collected for $5^{\circ} < 2\theta < 70^{\circ}$ by a Philips X'pert Pro MPD using the Bragg-Brentano geometry with Cu $K\alpha$ radiation. Because the antiferromagnetic ground state in CsVF₄ 78 79 was found to be broken by $H \ge 5 \text{ kOe}$ [12,13], a weaker magnetic field should be applied to α -RbCrF₄. 80 The temperature dependence of χ was measured using a superconducting quantum interference device magnetometer (Quantum Design, MPMS-XL) from 2 K to 350 K. In this study, we defined the magnetic 81 82 susceptibility as $\chi = M / H$. The FC and ZFC data were collected after applying H = 10 Oe and 1 kOe at T = 350 K and 2 K, respectively. Because the $\gamma(T)$ curves for the FC and ZFC data overlapped, the ZFC 83 84 data were omitted in the following discussion.

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3. EXPERIMENTAL RESULTS AND DISCUSSION

Figure 2(a) presents the sharp XRD peak profiles obtained after sintering at 640 °C. According to the study by Babel, the space group of aristotype α -RbCrF₄ is P4/*mmm*, and the lattice constants are

a = 3.719 Å and c = 6.442 Å [17]. However, as observed in the inset of Fig. 2(a), the splitting of the peaks 90 indicates the appearance of a superstructure beyond the wavelength difference between Cu $K\alpha_i$ and 91 Cu $K\alpha_{2}$. We believe that the crystal system is orthorhombic and that the lattice constants are 92 $2a \approx 2b \approx 7.348$ Å and c = 6.442 Å [16,17]. Based on the analysis of structural phase transition in $A^{I}M^{III}F_{4}$, 93 α -RbCrF₄ is expected to be found in the *P*mmn ($2a \times 2b \times c$) space group[20], with Aizu's 4/mmm*Fmmm* 94 notation indicating a ferroelastic state in the orthorhombic phase[4]. Although the superstructure has not 95 vet been determined, we denoted the fundamental peaks using the indices shown in Fig. 2(a) based on 96 97 the superstructure reported in Ref. 16.

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We determined the best sintering temperature to be 640 °C and then post-annealed the samples at 500 99 $^{\circ}$ C under HF gas; we denote this as the usual method[15]. As shown in Fig. 2(b), we measured the T 100 dependence of χ for α -RbCrF₄ at H = 10 Oe to prevent the saturation of impurity-induced weak 101 ferromagnetic moments at high magnetic fields. As observed in $\chi(T)$ data for the samples obtained by 102 103 the usual method, an anomaly indicating a weak ferromagnetic moment appeared at T = 15.0(5) K. When 104 we crystallized several samples under the same conditions, the transition temperatures T' remained 105 unchanged; however, the values of χ at 2.0 K varied widely among different batches. We believe that this magnetic transition at T' is due to extrinsic properties, i.e., the presence of some magnetic impurities. 106

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109 Figure 2: (a) X-ray diffraction pattern for α -RbCrF₄ after sintering at 640 °C. As shown in the 110 inset, XRD peak splitting that indicates a superstructure appeared beyond the wavelength difference between Cu $K\alpha_1$ and Cu $K\alpha_2$. Fundamental peaks are denoted by the indices on the 111 112 basis of the superstructure reported in Ref. 16. (b) Low-temperature magnetic susceptibility [$\chi(T)$] for samples obtained using the usual method (open circles) and the RS method (closed 113 circles). Both datasets agree well with each other above T'=15.0(5) K. However, an anomaly 114 indicating the weak ferromagnetic moment appeared in the data for the samples obtained by the 115 116 usual method, whereas no anomaly appeared in the data for the samples obtained by the RS 117 method.

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119 To further purify polycrystalline α -RbCrF₄, we improved the pretreatment of the sample before primary 120 sintering at 640 °C. Figure 3 presents the scheme of the improved pretreatment, i.e., processes 3-6 were repeated several times before primary sintering. Finally, the samples were heated at 640 °C for 50 h 121 122 under N₂ gas flow (200 cc/min) and then post-annealed at 500 °C under HF gas. We will denominate the 123 improved pretreatment "the return to synthetic precursor method" (abbreviated as the RS method). As 124 observed in Fig. 2(b), when we compare the data for the samples obtained using the RS method with those obtained using the usual method, the $\chi(T)$ curves for both methods agree well with each other 125 126 above 15 K; however, no anomaly appeared below 15 K in the data for the samples obtained by the RS 127 method. On the other hand, the samples obtained by the two methods are indistinguishable based on the 128 XRD data. In future, we will refine the superstructure for α -RbCrF₄ to obtain high-quality α -RbCrF₄ 129 where the magnetic phase transition at T' will be absent.



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131 Figure 3: Schematic of the improved pretreatment for α -RbCrF₄. Processes 3-6 were repeated 132 several times before primary sintering at 640 °C. We will denominate the improved pretreatment 133 "the return to synthetic precursor method" (abbreviated as the RS method).

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135 Figure 4(a) shows the $\chi(T)$ curves at H = 1 kOe over the entire temperature range for the high-quality 136 α -RbCrF₄. A broad $\chi(T)$ peak indicating a low-dimensional antiferromagnetic behavior appeared at 137 $T_{\rm max} \approx 47 \text{ K}$. As observed in the inset of Fig. 4(a), a sharp $\chi(T)$ peak appeared with the curvature similar to that of $CsVF_4$ at H = 200 Oe[12]. Therefore, we conclude that an antiferromagnetic phase 138 transition occurs at T_N = 29.3(2) K in α -RbCrF₄. Figure 4(b) shows the 1/ $\chi(T)$ curve. We fitted the 139 $1/\chi(T)$ data above 250 K to the Curie-Weiss law $[\chi(T) = C/(T - \Theta_{CW})$ and $C = N_A g^2 \mu_B J (J+1)/(3k_B)$]; the 140 resulting Weiss temperature (Θ_{cw}) and effective magnetic moment were -67(3) K and $3.98(3) \mu_{B}$, 141 respectively. The effective magnetic moment agrees with the spin-only value 3.87(3) $\mu_{\rm B}$ within the 142 143 experimental error. Applying the molecular field theory to solve the Hamiltonian given by

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$$H_{\text{cal}} = -2\sum_{\langle i,j \rangle} J_{i,j} \boldsymbol{S}_i \cdot \boldsymbol{S}_j , \quad (3.1)$$

145 the Weiss temperature is written as

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$$\Theta_{\rm CW} = \frac{2zJS(S+1)}{3k_{\rm B}},$$
 (3.2)

where *z* is the number of nearest neighbors and *J* is the isotropic exchange interaction, provided that this equation is valid for $T > |\Theta_{cw}|$. Applying $\Theta_{cw} = -67(3)$ K and z = 4, we estimated the intra-layer exchange interaction to be $J_{intra} / k_{B} = -6.7(3)$ K.

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Figure 4: (a) Temperature dependence of the magnetic susceptibility (χ) for the high-quality α -RbCrF₄. A broad $\chi(T)$ peak appeared at $T_{\text{max}} \approx 47 \text{ K}$. The solid line is the best fit of the theoretical formula reported by Lines in Ref. 21, which yields $J_{\text{intra}} / k_{\text{B}} = -6.6 \text{ K}$ and $\chi_{\text{const}} = -9.4 \times 10^{-5} \text{ emu/mol}$. The inset shows a sharp peak for $\chi(T)$, indicating an antiferromagnetic phase transition at $T_{\text{N}} =$ 29.3(2) K. (b) Temperature dependence of $1/\chi(T)$. The solid line is the Curie-Weiss law; the resulting Weiss temperature is $\Theta_{\text{CW}} = -67(3) \text{ K}$.

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160 For a quadratic-layer Heisenberg antiferromagnet, the magnetic susceptibility reported by Lines can be 161 expressed in the high-temperature region [$\chi_{2D}(T)$] by a series expansion[21]. To subtract the Tindependent term ($\chi_{\rm const}$), i.e., the diamagnetic contribution and Van Vleck term, we attempted to use 162 163 $[\chi_{2D}(T) + \chi_{const}]$ with the experimental data in the 200-350 K temperature range far above $|\Theta_{cw}|$. For the best-fit curve shown in Fig. 4(a), the values of J_{intra} and χ_{const} are determined to be -6.6(1) K and 164 $-9.4(1) \times 10^{-5}$ emu/mol, respectively. The values of J_{intra} obtained by the different analyses are in good 165 Using the relation 166 agreement. between $J_{\rm intra}$ and $T_{\rm max}$, we further obtained $J_{\text{intra}}/k_{\text{B}} = -T_{\text{max}}/[1.12 \times S(S+1) + 0.10] = -5.5(2) \text{ K}$ [21]. This value is smaller than the others. The broad 167 $\chi(T)$ peak in α -RbCrF₄ is necessary to re-consider not only the development of an antiferromagnetic 168

169 short-range order accompanied with the inter-layer exchange interaction (J_{inter}) but also the existence of 170 the inequivalent magnetic sites, i.e., the tilting of CrF₆ octahedra in the opposite manner[20]. Based on the theoretical investigations of the effect of J_{inter} by Ginsberg, $\chi_{2D}(T)$ must be modified at $T < |\Theta_{CW}|$, but 171 172 the value of T_{max} is almost invariant[22]. According to Yasudas' theoretical investigations of the relation between J_{inter} and T_{N} in both quantum and classical two-dimensional antiferromagnets, we obtained 173 $J_{\text{inter}}/J_{\text{intra}} \approx 0.05$ from Fig. 3 in Ref. 23, where we roughly estimated $(k_{\text{B}}T_{\text{N}})/[2]J_{\text{intra}}|S(S+1)] \approx 0.6$. Based 174 on the magnetic susceptibility and neutron diffraction experiments on CsVF₄, the magnetic ground state 175 176 may change from the antiferromagnetic state to some other state with ferromagnetic moments at a critical magnetic field (H_c)[12,13]. In α -RbCrF₄, we re-expressed H_c at T = 0 K as $|g_1 - g_2|\mu_B H_C = 6|J_{inter}|/k_B$, 177 where g_1 and g_2 correspond to the g-value of each inequivalent magnetic site[12]. Because 178 $|g_1 - g_2| \approx 0.01 - 0.05$ in Cr³⁺ compounds[24], we roughly estimated the critical field as 179 $H_{\rm C}[\rm kOe] \approx 2 \times 10^3 - 1 \times 10^4 |J_{\rm inter}|[\rm K]$. Consequently, we found that the $\chi(T)$ data at H = 1 kOe shown in Fig. 180 4 indicate the presence of an intrinsic antiferromagnetic ground state. Thus, the absence of the 181 differences between the FC and ZFC $\chi(T)$ curves is because of the absence of a ferromagnetic moment. 182

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184 4. CONCLUSIONS

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We successfully synthesized high-quality α -RbCrF₄, a quasi-two-dimensional Heisenberg 186 antiferromagnet with S = 3/2 using the RS method, which is an improved pretreatment method. The RS 187 method can be applied to many powdered polycrystalline or monocrystalline fluorides before primary 188 sintering. XRD experiments revealed that α -RbCrF₄ consists of a superstructure in the *ab*-plane due to 189 190 the distorted square lattice caused by the tilting of CrF₆ octahedra, as indicated by the appearance of splitting of the XRD peaks. However, the structure of α -RbCrF₄ was not determined. Magnetic 191 192 susceptibility experiments did not find any extrinsic anomaly due to impurities at T'=15.0(5) K. The intrinsic $\chi(T)$ curve exhibited a broad $\chi(T)$ peak at $T_{\text{max}} \approx 47 \text{ K}$ and showed the occurrence of an 193 antiferromagnetic phase transition at T_N = 29.3(2) K. Using the $\chi(T)$ data in the high-T region, we 194 195 obtained $J_{intra} / k_B = -6.6(1) \text{ K}$ and $J_{inter} / J_{intra} \approx 0.05$. As a result, the $\chi(T)$ data at H = 1 kOe indicate the 196 presence of an intrinsic antiferromagnetic ground state. If the degree of the good two-dimensionality is 197 increased, the antiferromagnetic ground state may change under low magnetic fields because of the competition between the Zeeman energy and J_{inter} . We expect that the singular curvature of the $\chi(T)$ 198 199 peak at T_N = 29.3 K is because of the existence of inequivalent magnetic sites, i.e., the different tilting schemes of the F⁻ octahedra surrounding the Cr³⁺ ions at different sites. In future, the superstructure will 200 be determined to clarify whether ferromagneticity, ferroelectricity, and ferroelasticity coexist in this 201 202 material. Furthermore, the temperature dependence of the heat capacity under high magnetic fields and 203 high field magnetization process will be measured to confirm the magnetic-field-induced phase transition. 204

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